

Fe-Ga-Sb (Iron-Gallium-Antimony)

V. Raghavan

Recently, Deputier et al. [2002Dep] determined an isothermal section for this system at 600 °C, which depicts an extensive solubility of Ga in the Fe₃Sb₂ (ϵ) binary phase.

is stoichiometric and has the cubic ZnS (sphalerite) type structure. The mutual solubility between Ga and Sb is very limited.

Binary Systems

The Fe-Ga system [1993Oka1] is characterized by the presence of a closed γ loop and several ordered forms of the bcc Fe based solid solution (α Fe). α' has the CsCl type ordered structure. The structure of α'' is not known. α''' has the cubic BiF₃ type structure. The intermediate phases of the system are: Fe₃Ga, Fe₆Ga₅, Fe₃Ga₄, and FeGa₃. The first two have high- and low- temperature modifications. For crystal structure data, see [1993Oka1] and [Pearson3]. The Fe-Sb phase diagram [1993Oka2] depicts two intermediate phases. The NiAs-type B8₁ phase Fe₃Sb₂ (ϵ) has a homogeneity range of 40-47 at.% Sb. The other intermediate phase FeSb₂ is stoichiometric and has orthorhombic symmetry. [Pearson3] lists two orthorhombic types for this phase: FeS₂-type (*Pnm*) and FeSb₂-type (*Pnn*2). The Ga-Sb phase diagram [1988Nga] is characterized by the presence of the congruently-melting intermediate phase GaSb. It

Ternary Isothermal Section

With starting materials of purity of $\geq 99.99\%$, [2002Dep] heated about 30 alloy compositions to 1000 °C for 72 h in evacuated silica tubes. The cooled samples were ground to powder and annealed at 600 °C for a long time to ensure homogeneity and then quenched in ice-water mixture. The phase equilibria were studied by x-ray powder diffraction and by energy dispersive spectroscopy attached to a scanning electron microscope (SEM-EDS). The isothermal section at 600 °C constructed by [2002Dep] is redrawn in Fig. 1 to agree with the accepted binary data. The ordered form of bcc Fe (α''') and the Sb-rich liquid, omitted by [2002Dep], are schematically indicated.

The main feature of the section in Fig. 1 is the large solubility of Ga in the Fe₃Sb₂ (ϵ) phase, with Ga substituting for Sb. The Ga-rich end has the composition range of Fe₇Ga_{0.8}Sb_{1.2} ($2.15 \leq t \leq 2.80$), that corresponds to an Fe

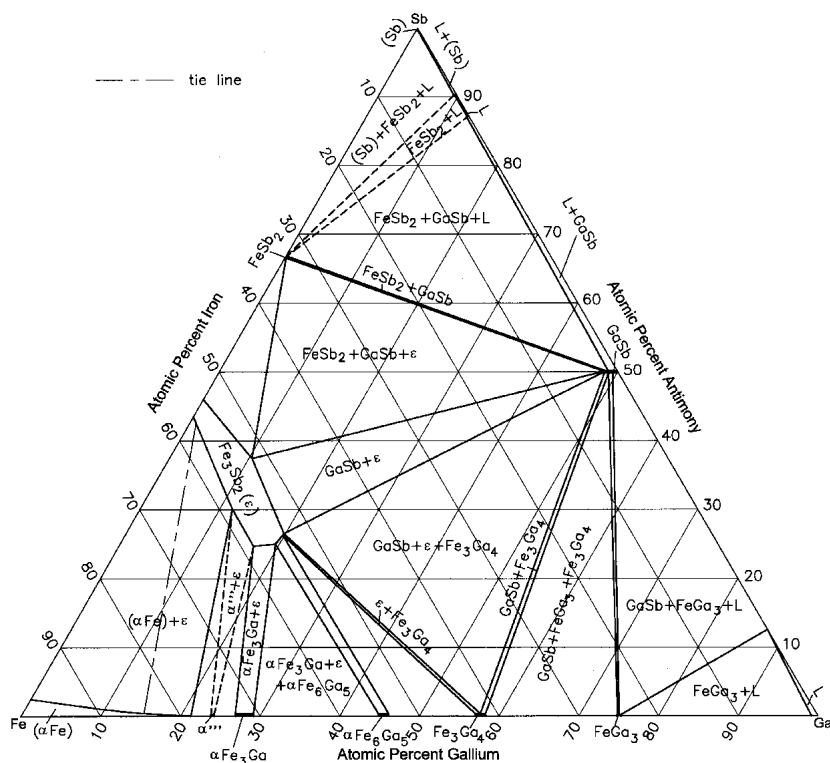


Fig. 1 Fe-Ga-Sb isothermal section at 600 °C [2002Dep]

Section II: Phase Diagram Evaluations

range of 52-58 at.%. This range does not cover the composition Fe_7GaSb found at 800 °C by [1994Moz]. $\text{Fe}_7\text{Ga}_{0.8}\text{Sb}_{1.2}$ has hexagonal symmetry and a structure intermediate between $B8_1$ and $B8_2$ types. No extra lines are seen in the x-ray powder patterns, indicating that Ga and Sb atoms are randomly distributed in the non-metal sublattice. In NiAs ($B8_1$), there are two types of interstitial sites which are vacant. In Ni_2In ($B8_2$), both the interstitial sites are occupied by Ni atoms. In $\text{Fe}_7\text{Ga}_{2-x}\text{Sb}_x$ (ϵ), the extra Fe atoms (as compared to the NiAs structure) are distributed randomly over the two interstitial sites. This phase can be considered either as a partially-filled NiAs type structure or a metal-deficient Ni_2In type structure [2002Dep].

In Fig. 1, in addition to $\text{Fe}_7\text{Ga}_{2-x}\text{Sb}_x$, GaSb forms tie lines with several other binary phases. This means that annealing at lower temperatures (<500 °C) a Fe/GaSb contact does not result in a stable, monophasic, epitaxial heterostructure as is found in the As-Fe-Ga system [2002Dep].

References

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